

# Theoretical investigation into optical and electronic properties of oligothiophene derivatives with phenyl ring as core or end-capped group in linear and V-shape

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**Abstract** A series of thiophene-based oligomers has been designed to explore their optical, electronic, and charge transport properties for charge transport materials. These oligomers consist of oligothiophene, oligo(thienylenevinylene), and *m*- or *p*-phenyl as the core in two shapes (linear shape and V-shape). Phenyl ring as the end-capped group is also investigated in the linear shape. The DFT-PBE0/6-31G(d,p) and the TD-PBE0/6-31+G(d,p) calculated results reported herein show that the V-shape oligomers have larger HOMO-LUMO gaps because of meta-substitutions on phenyl cores, corresponding to blue shifts of absorption spectra. The linear oligomers with phenyl ring as end-capped group display red shifts of absorption spectra. The V-shape oligomers provide small reorganization energies. Our recommended polymer possessing 1,2,4-phenyl core and longer OTV side fragments is a good candidate for the design of charge transport and/or solar cell materials.

**Keywords** Oligothiophene derivatives · Optical and electronic properties · Charge transport property

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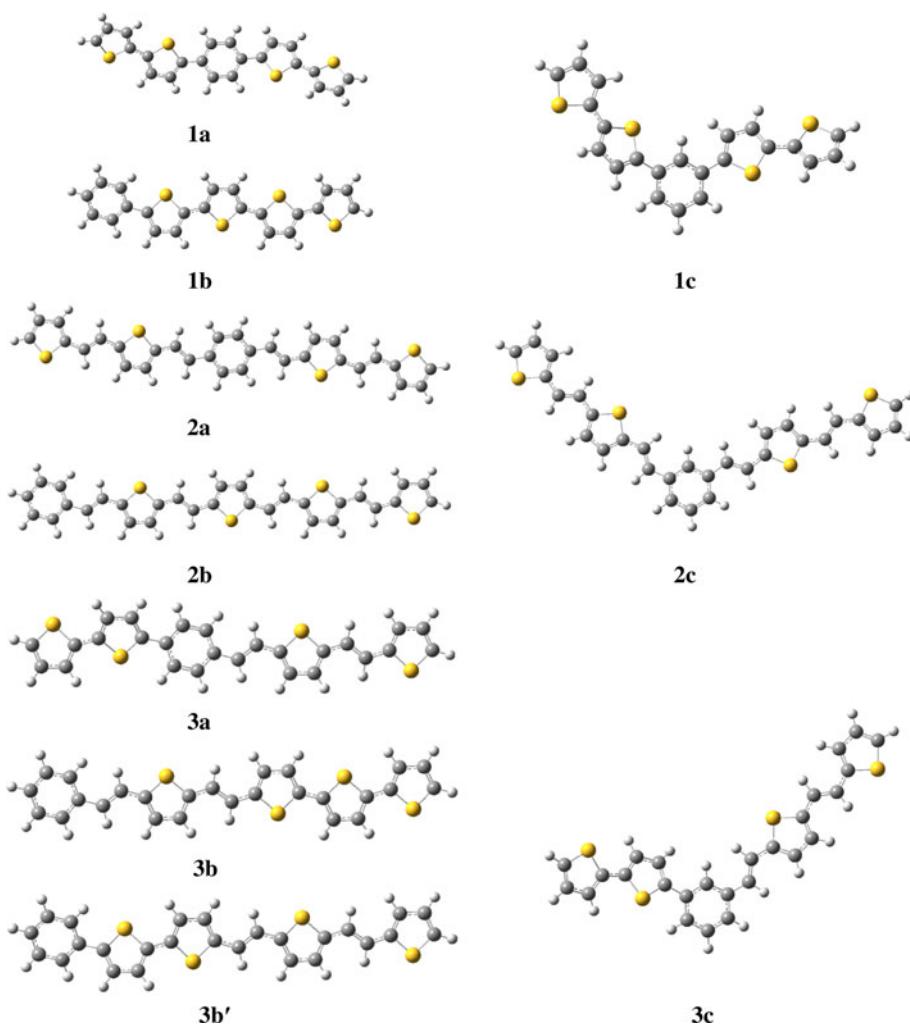
## 1 Introduction

Organic materials used as optical and electronic devices, such as organic light-emitting diodes (OLEDs), organic field-effect transistors (OFETs), and organic solar cells (OSCs), have recently received a great deal of attention from the standpoint of potential technological applications as well as fundamental science [1–24]. The devices using organic materials are attractive because of their excellent properties, for example, light weight, potentially low cost, capability of thin-film, large-area, and flexible device fabrication.

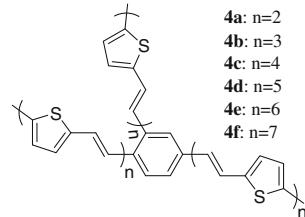
Functional thiophene-related oligomers and polymers have drawn much attention because they are the most frequently used  $\pi$ -conjugated systems among organic materials, particularly as active components in organic electronic devices and molecular electronics [25–28]. Many substituted derivatives have been designed and synthesized, and their optical and electronic properties have also been investigated [29–42]. Recently, the traditional linear systems have been vastly extended to higher dimensionalities and novel topologies [43]. It is well known that charge transport property in organic materials is one of the most important properties in the performance of OLEDs [44, 45], OFETs [46, 47], and OSCs [48–50]. Understanding the relationship between the two different molecular shapes and charge transport property of a material is a key factor for providing the guideline for device design, and great theoretical research efforts are currently being made in this regard [51–53].

With the aim to explore the detail of the two different molecular shapes for charge transport and/or solar cell applications of well-investigated oligothiophene (OT) and oligo(thienylenevinylene) (OTV), we designed three series

**Fig. 1** Molecular structures of the investigated molecules



of OT and OTV derivatives by introducing phenyl ring as core or end-capped group forming two topologies, i.e., linear or V-shape (see Fig. 1). For understanding the effects of the position of phenyl ring and the two different molecular shapes for OT derivatives on their optical and electronic properties, the first series is obtained as follows: (1) **1a** corresponds to the linear OT with *p*-phenyl as core; (2) **1b** refers to the linear OT with phenyl ring as end-capped group; (3) **1c** is the V-shape OT with *m*-phenyl as core. The second series (**2a–c**) is considered for investigating the influence of the position of phenyl ring and the two different molecular shapes of OTV derivatives on their optical and electronic properties. To study the properties for co-oligomers of oligothiophene-phenyl-oligo(thienylenevinylene), **3a–c** were designed. The frontier molecular orbitals (FMOs) including the highest occupied molecular orbital (HOMO) and the lowest unoccupied molecular orbital (LUMO) energies, the HOMO–LUMO gaps ( $E_g$ ) as well as the absorption spectra were predicted. According to the Marcus model [54–56], there are two key parameters for charge transfer rate which are electronic



**Scheme 1** Chemical structures of **4a–f** as the most promising candidates

coupling matrix element and reorganization energy, and the latter is more important [57]. Hence, we mainly study the reorganization energy for electron ( $\lambda_e$ ) and hole ( $\lambda_h$ ) of the designed molecules to investigate their charge transport properties. In addition, the correlations between these properties and their molecular topologies were also discussed. On the basis of detailed investigation of designed oligomers, the most promising candidates were designed as **4a–f** (Scheme 1); the various properties of corresponding polymer were predicted by extrapolation technique.

## 2 Computational details

All the calculations were carried out with the aid of Gaussian 03 package [58]. The density function theory (DFT) [59] with the PBE0 [60, 61] method using the 6-31G(d,p) [62–64] basis set was selected in all the geometry optimization including neutral, cation, and anion molecules. Our previous work [65] and the reports by Jacquemin [66–69] suggested that the PBE0 appeared notably adapted to sulfur-bearing molecules. Molecules **1a** and **2a** possess  $C_2$  symmetries. Considering the fact that oligothiophene is flat in crystal [70], **1a'–c'** are restricted in planar conformation in comparison with **1a–c**, respectively (see Fig. S1, Supporting Information). **1c** and **2c** are more stable ones in their possible conformations, respectively, (see Fig. S1 and Table S1, Supporting Information). The absorption spectra of all the compounds were investigated by the TD-PBE0/6-31+G(d,p) [64, 71–75] method based on the optimized geometries at the PBE0/6-31G(d,p) level. To investigate the solvent effects on the optical and electronic properties of the derivatives, we adopted the self-consistent reaction field (SCRF) approach with the polarizable continuum model (PCM) [62, 76–81] using benzene, acetonitrile, and chloroform solvents to calculate the absorption spectra of **1a** as representation. Recently, the B3LYP/6-31G(d,p) [82] functional was successfully used to calculate charge transport parameters for OT [83]. Moreover, in order to compare with the interested results reported previously [84, 85], the reorganization energies for electron ( $\lambda_e$ ) and hole ( $\lambda_h$ ) of the molecules were predicted from the single point energy at the B3LYP/6-31G(d,p) level on the basis of the PBE0/6-31G(d,p) optimized neutral, cationic, and anionic geometries. The reorganization energy can be divided into two parts, external reorganization energy ( $\lambda_{\text{ext}}$ ) and internal reorganization energy ( $\lambda_{\text{int}}$ ).  $\lambda_{\text{ext}}$  represents the effect of polarized medium on charge transfer; on the other hand,  $\lambda_{\text{int}}$  is a measure of structural change between ionic and neutral states [86]. In the solid-state film of charge transport and/or

solar cell materials,  $\lambda_{\text{ext}}$  is not expectable to be the main factor of reorganization energy, because of the low dielectric constant of medium [87]. Therefore, we focus the discussion on the internal reorganization energy of the isolated active organic  $\pi$ -conjugated systems due to ignoring any environmental relaxation and changes in this paper. Hence, the  $\lambda_e$  and  $\lambda_h$  values can be calculated by Eqs. 1 and 2 [88]:

$$\lambda_e = [E_0^- - E_-] + [E_-^0 - E_0] \quad (1)$$

$$\lambda_h = [E_0^+ - E_+] + [E_+^0 - E_0] \quad (2)$$

Where  $E_0^+$  ( $E_0^-$ ) is the energy of the cation (anion) calculated with the optimized structure of the neutral molecule. Similarly,  $E_+$  ( $E_-$ ) is the energy of the cation (anion) calculated with the optimized cation (anion) structure,  $E_+^0$  ( $E_-^0$ ) is the energy of the neutral molecule calculated at the cationic (anionic) state. Finally,  $E_0$  is the energy of the neutral molecule at the ground state. The B3LYP/6-31G(d,p) level was also used for the solvent reorganization energies of electron and hole for **1a** in benzene, chloroform, and acetonitrile solvents, respectively, using PCM models as representation. The density of states (DOS) was calculated and convoluted using the Gausssum 1.0 [89].

## 3 Results and discussion

### 3.1 Molecular structures

In general, the comparison of the optimization results for three fragments (thiophene, phenyl ring, and thienylenevinylene) of investigated molecules does not reveal any significant change in the geometry of the skeleton. The main structural changes occurred between the adjacent units, especially the geometrical parameters between phenyl ring and thiophene fragments. The molecular structures are presented in Scheme S1 (see the Supporting Information), and the calculated results have been listed in Table 1. For OT

**Table 1** Optimized geometrical parameters for the structures of molecules

Species	Geometrical parameters			
<b>1a</b>	$m_{2-7}$ (Å) 1.46	$m_{5-9}$ (Å) 1.46	$\alpha_{3-2-7-8}$ (°) 22	$\alpha_{6-5-9-10}$ (°) 22
<b>1b</b>	$m_{2-7}$ (Å) 1.46		$\alpha_{3-2-7-8}$ (°) 27	
<b>1c</b>	$m_{2-7}$ (Å) 1.46	$m_{6-9}$ (Å) 1.46	$\alpha_{3-2-7-8}$ (°) 28	$\alpha_{1-6-9-10}$ (°) 27
<b>2a</b>	$m_{2-7}$ (Å) 1.45	$m_{5-9}$ (Å) 1.45	$\alpha_{1-2-7-8}$ (°) 2	$\alpha_{4-5-9-10}$ (°) 2
<b>2b</b>	$m_{2-7}$ (Å) 1.46		$\alpha_{1-2-7-8}$ (°) 0	
<b>2c</b>	$m_{2-7}$ (Å) 1.46	$m_{6-9}$ (Å) 1.46	$\alpha_{1-2-7-8}$ (°) 2	$\alpha_{5-6-9-10}$ (°) 0
<b>3a</b>	$m_{2-7}$ (Å) 1.46	$m_{5-9}$ (Å) 1.45	$\alpha_{1-2-7-8}$ (°) 13	$\alpha_{6-5-9-10}$ (°) 0
<b>3b</b>	$m_{2-7}$ (Å) 1.46		$\alpha_{1-2-7-8}$ (°) 0	
<b>3b'</b>	$m_{2-7}$ (Å) 1.46		$\alpha_{3-2-7-8}$ (°) 27	
<b>3c</b>	$m_{2-7}$ (Å) 1.46	$m_{6-9}$ (Å) 1.46	$\alpha_{1-2-7-8}$ (°) 30	$\alpha_{1-6-9-10}$ (°) 12

Bond lengths ( $m$ ) and bond angles ( $\alpha$ ) are listed

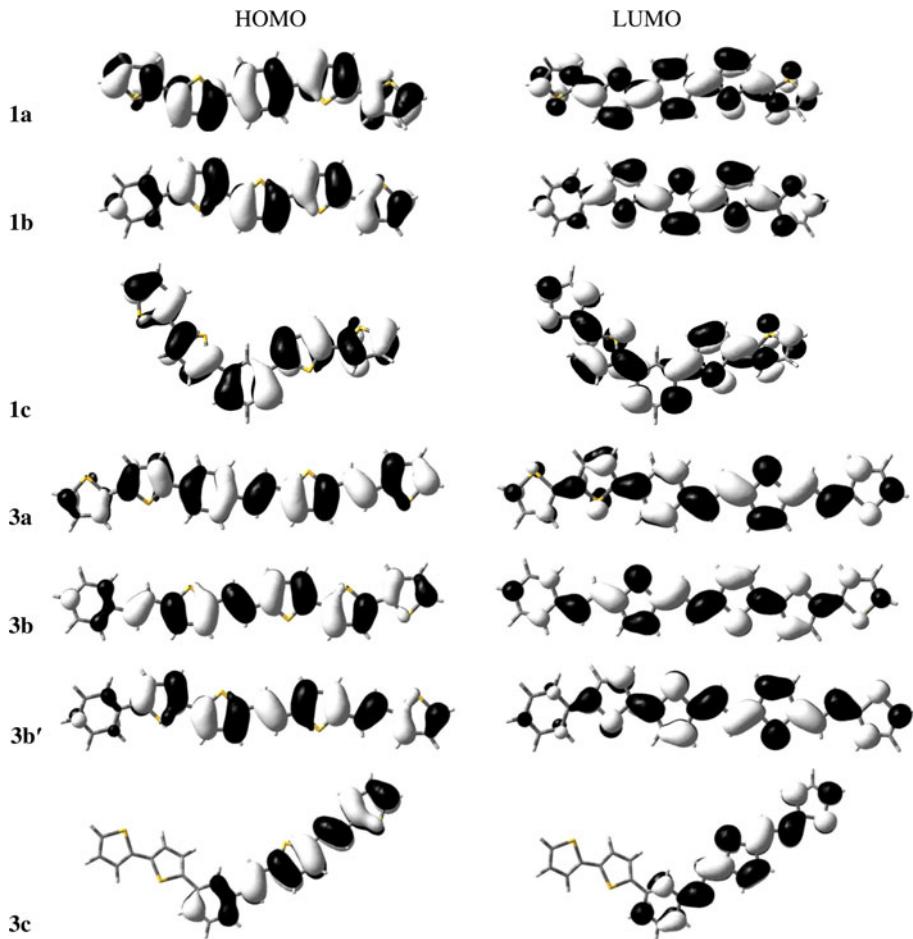
derivatives, both the two inter-ring distances between phenyl ring and thiophene fragments of **1a** are 1.46 Å. Its inter-ring dihedral angles are 22°, which is due to the steric hindrances. For **1b**, the inter-ring distances between phenyl ring and thiophene fragments are similar to those of **1a**, and its inter-ring dihedral angles are larger than those of **1a**. This indicates that different positions of phenyl ring have effects on the geometry parameters of OT derivatives. However, the inter-ring dihedral angles between phenyl ring and thiophene fragments of **1c** are increased to 28° and 27°, respectively, while the inter-ring instances have no obvious changes. It is because of larger steric hindrance in meta-substitution than that in para-substitution. The inter-ring distances of **1a'–c'** are similar (with the deviations less than 0.02 Å) to those of **1a–c**, respectively. For OTV derivatives, the inter-ring distances and dihedral angles between phenyl ring and thienylenevinylene fragments of **2a–c** are similar, which are about 1.45 Å and 2°, respectively. Therefore, the geometry parameters of OTV derivatives are only slightly influenced by the position of phenyl ring and the two different molecular shapes. For co-oligomers (**3a–c**), the inter-ring distances and dihedral angles between phenyl and

thiophene fragments are similar to those of **1a**, and the inter-ring distances and dihedral angles between phenyl and thienylenevinylene fragments are slightly larger than those of **2a**. Hence, the positions of phenyl ring and the two different molecular shapes have slight effects on the inter-ring distances for OT, OTV derivatives, and co-oligomers. While the inter-ring dihedral angles of OT and co-oligomer derivatives depend on their molecular shapes, V-shape molecules have larger inter-ring dihedral angles than linear ones (see Table 1) because of meta-substitution possessing larger steric hindrance than that of para-substitution.

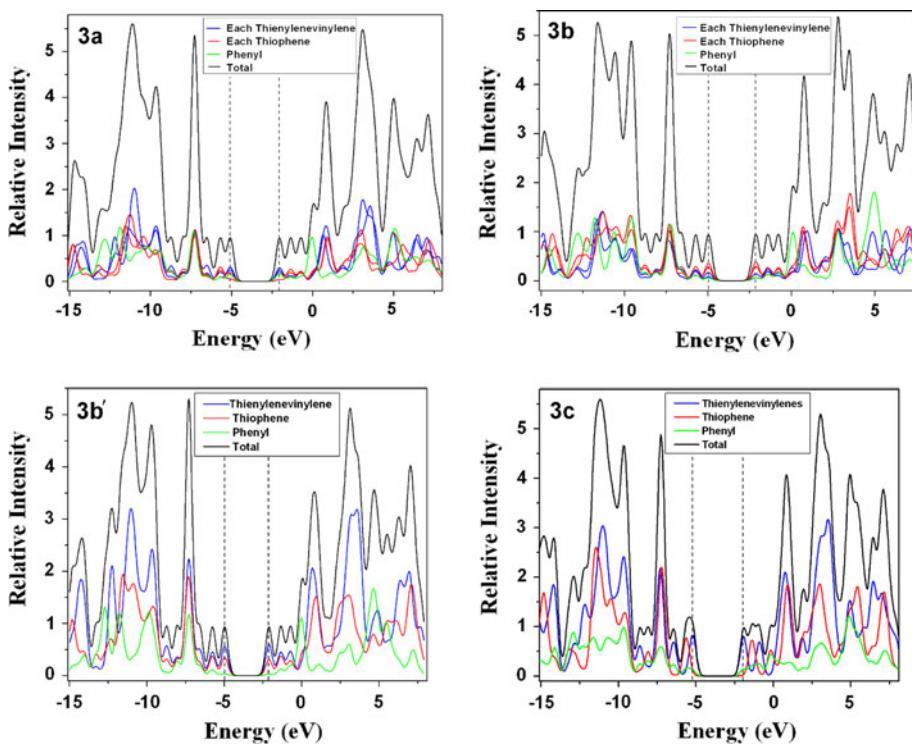
### 3.2 Frontier molecular orbitals

To characterize the optical transitions and the abilities of electron and hole transport, we calculated the distribution patterns of FMOs for all the investigated molecules (see Figs. 2 and S2). For a more detailed comparative study of the electronic structures, the total density of states (TDOS) and projected partial density of states (PDOS) on each fragment of the investigated molecules were calculated based on the current level of theory, as shown in Figs. 3 and S3.

**Fig. 2** The FMOs of the investigated molecules at the PBE0/6-31G(d,p) level



**Fig. 3** Total and partial density of states (TDOS and PDOS) around the HOMO–LUMO gap for investigated molecules (the value of full width at half maxima (FWHM) is 0.3 eV, dashed vertical lines indicate the HOMO and LUMO energies, respectively)



The origin of the geometric difference introduced by excitation can be explained, at least in qualitative terms, by analyzing the change in the bonding character of the orbitals involved in the electronic transition for each pair of bonded atoms [90]. As shown in Figs. 2 and S2, the ground state ( $S_0$ )  $\rightarrow$  first singlet excited state ( $S_1$ ) excitation process can be mainly assigned to the HOMO  $\rightarrow$  LUMO transition, which corresponds to a  $\pi-\pi^*$  excited singlet state. From Figs. 2 and S3, one can find that the FMOs of **1a** and **1c** are composed of contributions of the phenyl rings and thiophene fragments. For **1b**, the FMOs are highly delocalized throughout the thiophene fragments, with minor contributions from phenyl ring. The results of **1a'-c'** are similar to those of **1a-c** (see Figs. S2 and S3, Supporting Information). The FMOs of **2a** and **2c** are fairly distributed on the phenyl ring and thienylenevinylene fragments. For **2b**, the contributions from phenyl ring are smaller than those of thienylenevinylene fragments. These results reveal that the distribution patterns of FMOs for OT and OTV derivatives are influenced slightly by the position of phenyl ring and the two different molecular shapes. The contributions of HOMO and LUMO are only minor from one thiophene fragment (5% for HOMO and 4% for LUMO) for **3a**. Its phenyl ring (15% for HOMO and 19% for LUMO) and the other thiophene fragment (20% for HOMO and 20% for LUMO) offer small contributions, and two thienylenevinylene fragments have large contributions (60% for HOMO and 57% for LUMO). For **3b**, the contributions of phenyl ring and thienylenevinylene fragments

in FMOs are smaller than those of **3a**, and the contributions from thiophene fragments are larger than those of **3a**, respectively. For **3b'**, the contributions of phenyl ring in FMOs are smaller than those for **3a**, whereas the contributions coming from thienylenevinylene fragments are similar to those of **3a**, and the contributions by thiophene fragments are larger than those of **3a** in FMOs, respectively. These results show that the distribution patterns of FMOs are influenced slightly by the position of phenyl ring for linear co-oligomers. It is worth noting that the FMOs of **3c** have nearly no contributions from the thiophene fragments. The contributions from the phenyl ring are small in FMOs of **3c** (11% for HOMO and 15% for LUMO), and the contributions from thienylenevinylene fragments are dominant (88% for HOMO and 85% for LUMO). Therefore, various molecular topologies should influence significantly the distribution patterns of FMOs for co-oligomers.

For a better understanding of the effects of the position of phenyl ring and the two different molecular shapes on the energies of FMOs, we calculated HOMO and LUMO energies ( $E_{\text{HOMO}}$  and  $E_{\text{LUMO}}$ ), and the evaluations are schematically shown in Fig. S4 (see the Supporting Information). The calculated results reveal that for OT derivatives, the V-shape OT derivative (**1c**) has lower  $E_{\text{HOMO}}$  and higher  $E_{\text{LUMO}}$  and  $E_g$  values than those of linear ones (**1a** and **1b**). The similar tendency can be also found for **1a'-c'** (see Table S2, Supporting Information). As shown in Fig. S4, OTV derivatives with phenyl ring as core (**2a** and **2c**)

have larger  $E_g$  values than that of the one with phenyl ring as end-capped group (**2b**). For co-oligomers, the molecules with phenyl ring as end-capped group (**3b** and **3b'**) have higher  $E_{\text{HOMO}}$  and lower  $E_{\text{LUMO}}$  and  $E_g$  values than those with phenyl ring as core (**3a** and **3c**). As a result, the stabilization of the HOMO or LUMO depends on the position of phenyl ring and the two different molecular shapes, and the linear OTV with phenyl ring as end-capped group results in the smallest  $E_g$  value. The V-shape molecules have higher ionization potential (IP) and lower electron affinity (EA) values than those corresponding linear ones (see Table S3, Supporting Information), which are in line with the corresponding orders of absolute values for FMOs, respectively, as shown in Fig. S4 (see the Supporting Information).

### 3.3 Absorption spectra

The longest wavelength of maximal absorption ( $\lambda_{\text{max}}$ ) and corresponding oscillator strength ( $f$ ) of the investigated molecules at the TD-PBE0/6-31+G(d,p) level are listed in Table 2. For OT derivatives, the calculated  $\lambda_{\text{max}}$  values show the increasing order of **1c** < **1a** < **1b**, which is in good accordance with the corresponding reverse order of  $E_g$  values as shown in Sect. 3.2. A good agreement can be observed between our calculated  $\lambda_{\text{max}}$  value (414.81 nm) of **1a** and experimental observation (401.00 nm) [91]. This reveals that the level of theory we selected is reasonable for this kind of system. The predicted  $\lambda_{\text{max}}$  values of **1a** in different solvents are shown in Table S4 (see the Supporting Information). One can find that the deviations between

**Table 2** A comparative study of the predicted  $\lambda_{\text{max}}$  (nm) and corresponding  $f$  values for investigated molecules obtained at the TD-PBE0/6-31+G(d,p)//PBE0/6-31G(d,p) level and available experimental datum

Species	Main assignment	$\lambda_{\text{max}}$	$f$	Exp <sup>a</sup>
<b>1a</b>	H → L (0.65)	414.81	1.59	401.00
<b>1b</b>	H → L (0.65)	442.46	1.52	
<b>1c</b>	H – 1 → L (0.50)	358.01	1.29	
	H → L (0.15)			
	H → L + 1 (0.40)			
<b>2a</b>	H → L (0.64)	531.18	3.27	
<b>2b</b>	H → L (0.63)	557.51	3.24	
<b>2c</b>	H – 1 → L (0.49)	453.07	2.78	
	H → L + 1 (0.41)			
<b>3a</b>	H → L (0.64)	477.93	2.52	
<b>3b</b>	H → L (0.63)	512.99	2.42	
<b>3b'</b>	H → L (0.63)	505.94	2.45	
<b>3c</b>	H → L (0.62)	432.92	1.96	
	H – 1 → L + 1 (0.12)			

<sup>a</sup> Experimental datum of **1a** from Ref. [91]

different solvents are less than 16 nm. Therefore, different solvents (polar and non-polar) at PCM level have only slight effects on  $\lambda_{\text{max}}$  values. The order of corresponding  $f$  values is **1c** < **1b** < **1a**. The  $\lambda_{\text{max}}$  values of **1a'–c'** are predicted in the increasing order of **1c' < 1a' < 1b'**, and their corresponding  $f$  values are in sequence of **1c' < 1b' < 1a'** (see Table S2, Supporting Information). The  $\lambda_{\text{max}}$  value of **1a'/1b'/1c'** is larger than that of **1a/1b/1c**, respectively, due to its corresponding smaller  $E_g$  value. This indicates that OT derivatives will have large  $\lambda_{\text{max}}$  values in solid states due to their flat structures. For OTV derivatives, the  $\lambda_{\text{max}}$  values of **2a–c** are in the increasing order of **2c < 2a < 2b**, and their corresponding  $f$  values are in the sequence of **2c < 2b < 2a**. It suggests that OTV derivatives with phenyl ring as core (**2a** and **2c**) have smaller  $\lambda_{\text{max}}$  values than the one with phenyl ring as end-capped group (**2b**). The  $\lambda_{\text{max}}$  value of **2a/2b/2c** is larger than that of **1a/1b/1c** due to the smaller  $E_g$  value for **2a/2b/2c** in comparison with **1a/1b/1c**, respectively. For co-oligomers, the  $\lambda_{\text{max}}$  values are in the sequence of **3c < 3a < 3b' < 3b**, their predicted  $f$  values in absorption spectra are found in increasing order of **3c < 3b < 3b' < 3a**. These results show that the co-oligomers with phenyl ring as end-capped group (**3b** and **3b'**) have larger  $\lambda_{\text{max}}$  values than those possessing phenyl ring as core (**3a** and **3c**). The  $\lambda_{\text{max}}$  value of **3b** is close to that of **3b'** because of their similar distribution patterns of FMOs (Fig. 2). Furthermore, the V-shape co-oligomer (**3c**) has smaller  $\lambda_{\text{max}}$  value than those of linear ones. The  $\lambda_{\text{max}}$  and corresponding  $f$  values of co-oligomers are in between of those for corresponding OT and OTV derivatives, respectively. The calculated results reveal that V-shape molecules have smaller  $\lambda_{\text{max}}$  and corresponding  $f$  values than those linear shape molecules. This is ascribed to the meta-substitution inducing poor conjugation in comparison with para-substitution, which leads to V-shape molecules having larger  $E_g$ , smaller  $\lambda_{\text{max}}$ , and corresponding  $f$  values than linear ones. This result is in good agreement with earlier experimental observations [92, 93] and theoretical investigation (at the B3LYP/6-31G(d) level) [87] for similar systems. Moreover, the linear OTV derivative with phenyl ring as end-capped group (**2b**) has the largest  $\lambda_{\text{max}}$  value possessing more intensive spectrum.

### 3.4 Charge transport properties

Understanding the relationship between the two different molecular shapes and charge transport property of materials is a key point in providing good candidates for the design of charge transport and/or solar cell materials. The lower the reorganization energy value is, the higher the charge transfer rate is [94]. The calculated results for hole and electron according to Eqs. 1 and 2 are summarized in Table 3.

**Table 3** Calculated molecular  $\lambda_e$  and  $\lambda_h$  (eV) values

Species	$\lambda_e$	$\lambda_h$
<b>1a</b>	0.328	0.340
<b>1b</b>	0.344	0.332
<b>1c</b>	0.268	0.239
<b>2a</b>	0.228	0.234
<b>2b</b>	0.261	0.271
<b>2c</b>	0.152	0.161
<b>3a</b>	0.276	0.288
<b>3b</b>	0.259	0.272
<b>3b'</b>	0.299	0.293
<b>3c</b>	0.276	0.261

The calculated internal reorganization energies of **1a** for electron ( $\lambda_e$ ) and hole ( $\lambda_h$ ) are 0.328 and 0.340 eV in gas, respectively (see Table 3). The corresponding  $\lambda_e$  and  $\lambda_h$  values of **1a** in various solvents are shown in Table S5 (see the Supporting Information). It shows that the deviations of  $\lambda_e$  and  $\lambda_h$  values for **1a** in different solvents are in the range of 0.013–0.048 eV and 0.007–0.037 eV, respectively. Therefore, different solvents (both polar and non-polar) at PCM level used in this study only slightly affect the reorganization energies, i.e., the reorganization energies are solvent independent. Similar performance has been reported previously [95]. Thus, in this work, the solvent effect for reorganization energy has been neglected. For OT derivatives, the calculated  $\lambda_e$  and  $\lambda_h$  values of **1a** are close to those of **1b**, respectively. Thus, different positions of phenyl rings should have slight effects on the charge transfer rates for linear OT derivatives in the same environment. However, the  $\lambda_e$  and  $\lambda_h$  values of **1c** are much lower than those of **1a** and **1b**, respectively. We explain this result by following the two facts: the effective overall overlap of HOMO and LUMO for **1a** and **1b** are smaller than that of **1c** (see Fig. 2) resulting in smaller  $\lambda_e$  and  $\lambda_h$  values of **1c** than those of **1a** and **1b** [96]; the smaller geometry relaxation between ions and neutral geometrical structures occurs in the phenyl core for **1c** than that for **1a** and **1b** [97] (see Scheme S1 and Table S6, Supporting Information), respectively. For example, for **1c**, the geometrical structure differences between the cation and neutral one in the bond lengths of the phenyl groups are in the range of 0–0.01 Å. In the case of **1a** and **1b**, a range of 0.01–0.03 Å is observed. Therefore, the intrinsic charge transport properties for investigated OT derivatives should be molecular shape dependent, namely the V-shape OT derivative with *m*-phenyl core (**1c**) should have higher charge transfer rate than that of linear ones (**1a** and **1b**) in the same environment. The  $\lambda_e$  values are 0.224, 0.243, and 0.159 eV for **1a'–c'**, and their  $\lambda_h$  values are 0.250, 0.263, and 0.174 eV, respectively. The  $\lambda_e$  and  $\lambda_h$  values of **1a'–c'**

are smaller than those of **1a–c**, respectively. Thus, OT derivatives should have higher charge transfer rates in solid states because of their flat structures (considering the same environment). For OTV derivatives, the  $\lambda_e$  and  $\lambda_h$  values of **2a–c** are in the increasing sequence of **2c** < **2a** < **2b**. Moreover, the difference of  $\lambda_e$  values between **2a** and **2b** is 0.033 eV, and the difference of their  $\lambda_h$  values is 0.037 eV. The  $\lambda_e$  and  $\lambda_h$  values of **2c** are much lower than those of **2a** and **2b**, respectively, due to the same reasons as in **1a–c**. These results indicate that the charge transfer rates of OTV derivatives in linear shape are influenced slightly by the position of phenyl ring ignoring any environmental relaxation and changes. However, the different molecular topologies have significant effects on the charge transfer rates of OTV derivatives, that is, the V-shape OTV derivative (**2c**) should provide higher charge transfer rate than those of linear ones (**2a** and **2b**) in the same environment. The  $\lambda_e$  and  $\lambda_h$  values of **2a/2b/2c** are smaller than those of **1a/1b/1c**, respectively, implying OTV derivatives possessing higher charge transfer rates than those of OT derivatives considering the same environment. For co-oligomers, the calculated  $\lambda_e$  and  $\lambda_h$  values of **3a–c** are similar, indicating the position of phenyl ring and the two different molecular shapes having only a slight influence on  $\lambda_e$  and  $\lambda_h$  values. The calculated values of  $\lambda_e$  for **1c**, **2a–c**, and **3b** are lower than that of tris(8-hydroxyquinolino) aluminum (III) (Alq3) ( $\lambda_e$  = 0.276 eV) [84] which is a typical electron transport material. This implies that their electron transfer rates might be higher than that of Alq3. Meanwhile, the values of  $\lambda_h$  for **1c**, **2a–c**, **3a**, **3b**, and **3c** are lower than that of *N,N'*-diphenyl-*N,N'*-bis(3-methylphenyl)-(1,1'-biphenyl)-4,4'-diamine (TPD) ( $\lambda_h$  = 0.290 eV) [85], which is a typical hole transport material. This indicates that their hole transfer rates might be higher than that of TPD. In addition, the differences between  $\lambda_e$  and  $\lambda_h$  for the investigated molecules do not exceed 0.029 eV, implying better equilibrium properties for hole- and electron-transport. Therefore, these molecules can be used as good candidates for ambipolar charge transport materials under the proper operating conditions. It is worth mentioning that the V-shape OTV derivative with *m*-phenyl core (**2c**) shows the lowest  $\lambda_e$  and  $\lambda_h$  values among the investigated molecules. Hence, it is the best candidate for charge transport materials among the investigated molecules.

### 3.5 The properties of corresponding polymer based on appreciated oligomers by extrapolation

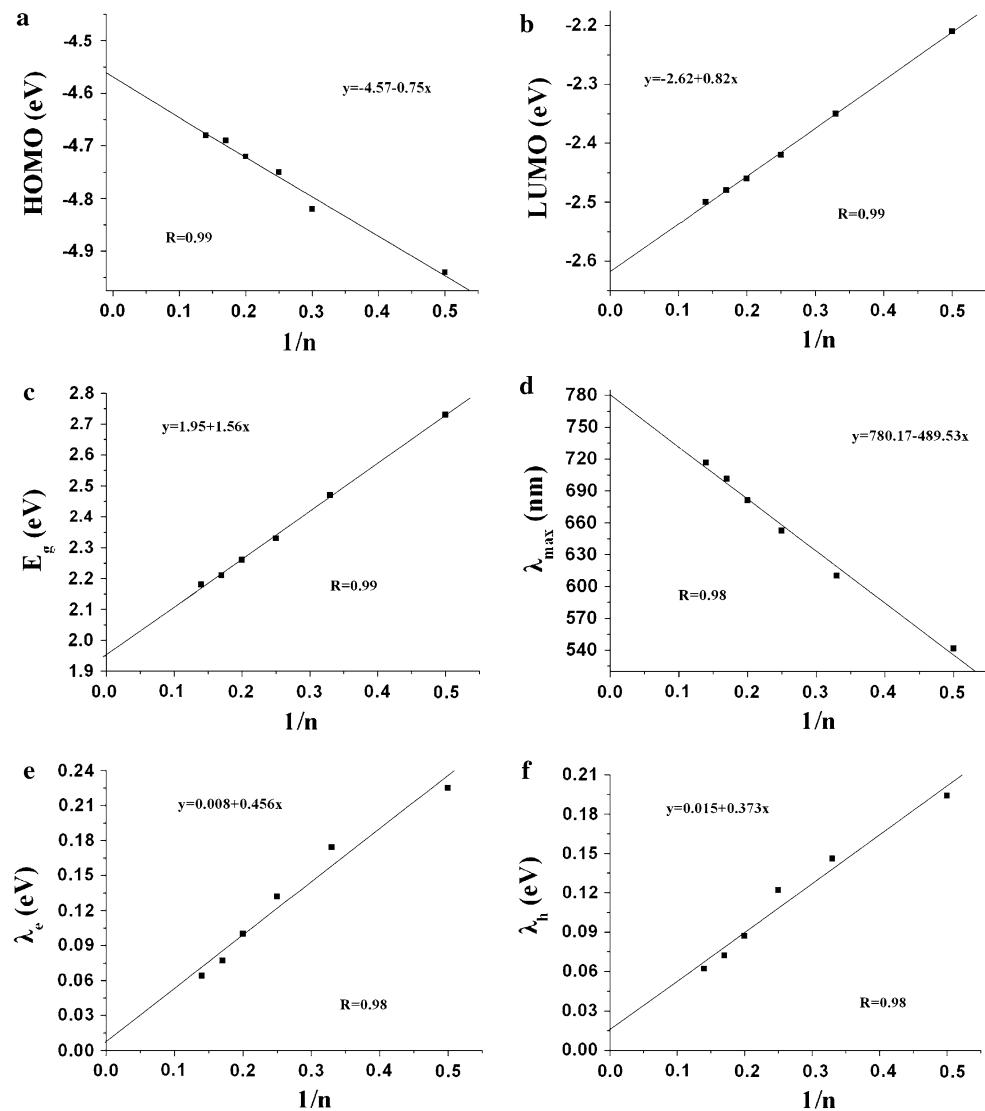
Considering the optical and electric properties for designed molecules above, the rational way to designing molecules toward charge transport materials should possess 1,2,4-phenyl core and OTV side fragments (**4a–f**, as shown in Scheme 1). The *m*-phenyl contributes to higher charge

transfer rate considering the same environment; meanwhile, *p*-phenyl as well as OTV fragments contribute to the better optical property. To predict the optical and electronic properties of corresponding polymer, we studied oligomers **4a–f** (Scheme 1). The HOMO, LUMO,  $E_g$ ,  $\lambda_{\max}$ ,  $\lambda_e$ , and  $\lambda_h$  values of the corresponding polymer were investigated by the extrapolation technique [98], which has been successfully employed to investigate several series of polymers [38, 39, 99–105]. Figure 4 presents the plots of HOMO, LUMO,  $E_g$ ,  $\lambda_{\max}$ ,  $\lambda_e$ , and  $\lambda_h$  values for **4a–f** as functions of reciprocal chain length for their corresponding oligomers, with assumed linear extrapolation to infinite chain length. Good linear relationships are found. The corresponding values are listed in Tables S7 and S8 (see the Supporting Information). The extrapolated HOMO, LUMO,  $E_g$ ,  $\lambda_{\max}$ ,  $\lambda_e$ , and  $\lambda_h$  values to the infinite chain length are –4.57, –2.62, 1.95 eV, 780.17 nm, 0.008, and 0.015 eV, respectively. These results reveal that increasing the conjugation

length of side fragments results in higher HOMO and lower LUMO energies, smaller  $E_g$ ,  $\lambda_e$ ,  $\lambda_h$ , and larger  $\lambda_{\max}$  values. This situation was also found in earlier theoretical investigation [106].

Polythiophene is one of the best  $\pi$ -conjugation systems in organic electronic devices and molecular electronics as mentioned above; thus, the  $E_g$ ,  $\lambda_{\max}$ ,  $\lambda_e$ , and  $\lambda_h$  values of corresponding polymer of oligothiophene (see Scheme S2, Supporting Information) were also calculated by extrapolation for comparison. The results are shown in Fig. S5 (see the Supporting Information). The predicted  $E_g$ ,  $\lambda_{\max}$ ,  $\lambda_e$ , and  $\lambda_h$  values of polythiophene are 2.11 eV, 529.14 nm, 0.173, and 0.212 eV, respectively. In comparison with the corresponding results of polythiophene, our designed polymer has smaller  $E_g$ ,  $\lambda_e$ ,  $\lambda_h$ , and larger  $\lambda_{\max}$  values than those of polythiophene, suggesting it to be a good candidate in solar cell, due to its narrower band gap, broader absorption region, and higher charge transfer rate.

**Fig. 4** **a** HOMO; **b** LUMO; **c**  $E_g$ ; **d**  $\lambda_{\max}$ ; **e**  $\lambda_e$ ; **f**  $\lambda_h$  as a function of  $1/n$  in **4a–f**, where  $n$  is the number of the repeating units along the polymer chain



#### 4 Conclusions

In the present work, we predicted the optical and electronic properties for a number of OT and OTV derivatives. They were investigated by means of quantum chemical method on the basis of the DFT-PBE0/6-31G(d,p) method. These results show that the two different molecular shapes play a key role in changing the FMOs energies. The V-shape OT derivative with meta-substituted phenyl ring will increase the  $E_g$  value because of the meta-substitution resulting in poor conjugation. The absorption spectra were evaluated at the TD-PBE0/6-31+G(d,p) level. Among the investigated molecules **1a–3c**, linear OTV derivative with phenyl ring as end-capped group (**2b**) owns the largest  $\lambda_{max}$  value (about 557.51 nm), which is in good agreement with the results of FMOs energies. The reorganization energies of the derivatives were also investigated on the basis of the B3LYP/6-31G(d,p) energies. The V-shape OTV derivative with meta-substituted phenyl as core (**2c**) provides the smallest reorganization energy among the investigated molecules **1a–3c**. Additionally, all the investigated molecules have better hole- and electron-transporting balance and can act as nice ambipolar materials. Using theoretical methodologies, it is possible to predict reasonable optical and electronic properties of the OT and OTV derivatives. On the basis of investigated results, we proposed a rational way for the design of charge transport and/or solar cell materials that the promising candidates should possess 1,2,4-phenyl core and longer OTV side fragments. The calculated results by extrapolation reveal that our designed polymer will possess better optical and electric properties than the well-known polythiophene. This study should be helpful in further theoretical investigations on such kind of systems and also to the experimental study for charge transport and/or solar cell materials.

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